

Alla Arakcheeva,^a Gervais Chapuis,^{a*} Václav Petricek^b and Vladimir Morozov^c

^aEcole Polytechnique Fédérale de Lausanne, Laboratoire de cristallographie, BSP, CH-1015 Lausanne, Switzerland, ^bInstitute of Physics, Academy of Sciences of the Czech Republic, Na Slovance 2, 182 21 Praha 8, Czech Republic, and ^cDepartment of Chemistry, Moscow State University, 119899 Moscow, Russia

Correspondence e-mail:
gervais.chapuis@epfl.ch

The role of second coordination-sphere interactions in incommensurately modulated structures, using β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$ as an example

Received 14 April 2005

Accepted 31 May 2005

The incommensurate palmierite-like structure of β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$, potassium ytterbium tetramolydate, has been refined in the (3 + 1)-dimensional monoclinic superspace group $X2/m(0\rho 0)00$, with $X = [0\ 0\ 0\ 0; \frac{1}{2}\ \frac{1}{2}\ 0\ 0; 0\ 0\ \frac{1}{2}\ \frac{1}{2}; \frac{1}{2}\ \frac{1}{2}\ \frac{1}{2}\ \frac{1}{2}]$ and the unit-cell parameters $a = 10.4054$ (16), $b = 6.1157$ (12), $c = 19.7751$ (18) Å, $\beta = 136.625$ (10)°; $\mathbf{q} = 0.6354$ (30) \mathbf{b}^* . The occupations of the K and Yb atomic positions are described by crenel functions. The structure model reveals a balanced interaction between the atoms of the first and second coordination spheres. It is shown that the third coordination sphere should not be neglected in studies of modulated structures. The ordering of the K and Yb atoms appears to be the driving force for the modulation of all the other atoms.

1. Introduction

The origin of the incommensurability in crystals was at the center of many theoretical models, which have been extensively discussed in the literature. In particular, ANNNI, DIFFOUR and other models (see, for example, Selke & Fischer, 1979; Bak, 1982; Janssen & Tjon, 1981) are based on competing interactions between nearest-neighbor and next-nearest-neighbor particles. From the theoretical considerations of Parlinski & Chapuis (1993) it was also shown that the role of atomic interactions not only of the first but also of the second coordination sphere (cation–cation for the anion-containing compounds) was fundamental for the formation of incommensurately modulated structures. The study of incommensurate structures as a probe to reveal atomic interactions in crystals has already been discussed with the examples of Na_2CO_3 and $\text{K}_3\text{In}(\text{PO}_4)_2$ (Chapuis & Arakcheeva, 2004; Arakcheeva *et al.*, 2003). In both crystals, the cation–cation interactions were identified as the driving forces and primary parameters of the modulations in the corresponding structures. Another example of cation–cation interactions was described by Dusek *et al.* (2002) for the monophosphate tungsten bronze $\text{K}_{1.33}\text{P}_4\text{W}_8\text{O}_{32}$, where the phase transition mechanism was associated with the release of internal strain between the K atoms and the surrounding PO_4 groups.

In the present paper we pursue our main goal of identifying the fundamental interactions in incommensurate structures with the investigation of β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$. Three modifications of the complex molybdenum oxide $\text{K}_5\text{Yb}(\text{MoO}_4)_4$ belonging to the palmierite [$\text{K}_2\text{Pb}(\text{SO}_4)$] structure type were recently identified by Morozov *et al.* (2003):

(i) the high-temperature phase α , space group $R\bar{3}m$, $a = 6.0372$ (1), $c = 20.4045$ (2) Å;

(ii) the low-temperature phase γ , space group $C2/c$, $a = 14.8236$ (1), $b = 12.1293$ (1), $c = 10.5151$ (1) Å, $\gamma = 114.559$ (1)°;

(iii) an intermediate-temperature phase β , which is incommensurate according to electron diffraction investigations.

The palmierite structure type is well known and the rhombohedral α - $K_5Yb(MoO_4)_4$ modification is as shown in Fig. 1. The structure is built up from two alternating layers along the trigonal c axis. The first layer consists of MoO_4 tetrahedra and MO_6 octahedra ($M = K, Yb$), whereas the second layer consists of KO_{10} polyhedra. In this article, however, we would like to single out another building unit which will play an important role in the description of the incommensurate structure, namely layer L as indicated in Fig. 1. This layer (Fig. 1*b*) is normal to the trigonal c axis and consists of MoO_4 tetrahedra, MO_6 octahedra and K atoms.

The whole structure thus consists of three identical layers related by the translation vectors of the R lattice. In the same publication (Morozov *et al.*, 2003), the crystal structures of phases α and γ were also refined. According to this study, the M position of the γ modification (Fig. 1) was statistically occupied by $(K_{0.5}Yb_{0.5})$, whereas ordering of K and Yb atoms was observed in the γ phase. A model based on a combination of α - and γ -structural entities was proposed for the incommensurate modification β .

The purpose of the present publication is the determination and refinement of the incommensurately modulated β phase. In addition, we analysed the atomic interactions resulting from the first (cation–oxygen) and second (cation–cation) coordination spheres in an attempt to reveal the complementary character of the two types of interactions.

2. Experimental study of the incommensurate β - $K_5Yb(MoO_4)_4$ structure

2.1. Preparation

Details of the preparation of $K_5Yb(MoO_4)_4$ by a conventional ceramic technique, as well as the conditions for the α , β and γ transformations and phase identification by electron- and X-ray powder diffraction techniques have previously been described by Morozov *et al.* (2003). The low-temperature modification γ was obtained at 893 ± 10 K. Annealing of this phase at 960 ± 10 K (close to the melting point) for 3 h followed by quick quenching by liquid nitrogen freezes the high-temperature α modification. Slow cooling from 960 K to room temperature induces the formation of the β modification.

Single crystals of β - $K_5Yb(MoO_4)_4$ were grown by spontaneous crystallization of the melt. The temperature was automatically controlled with an accuracy of ± 0.5 K. The cooling rate from 973 to 673 K was $3\text{--}5$ K h^{-1} , which was followed by cooling to room temperature in the cooling furnace regime.

2.2. X-ray experiment and results of structure refinement

The experimental details are summarized in Table 1. All

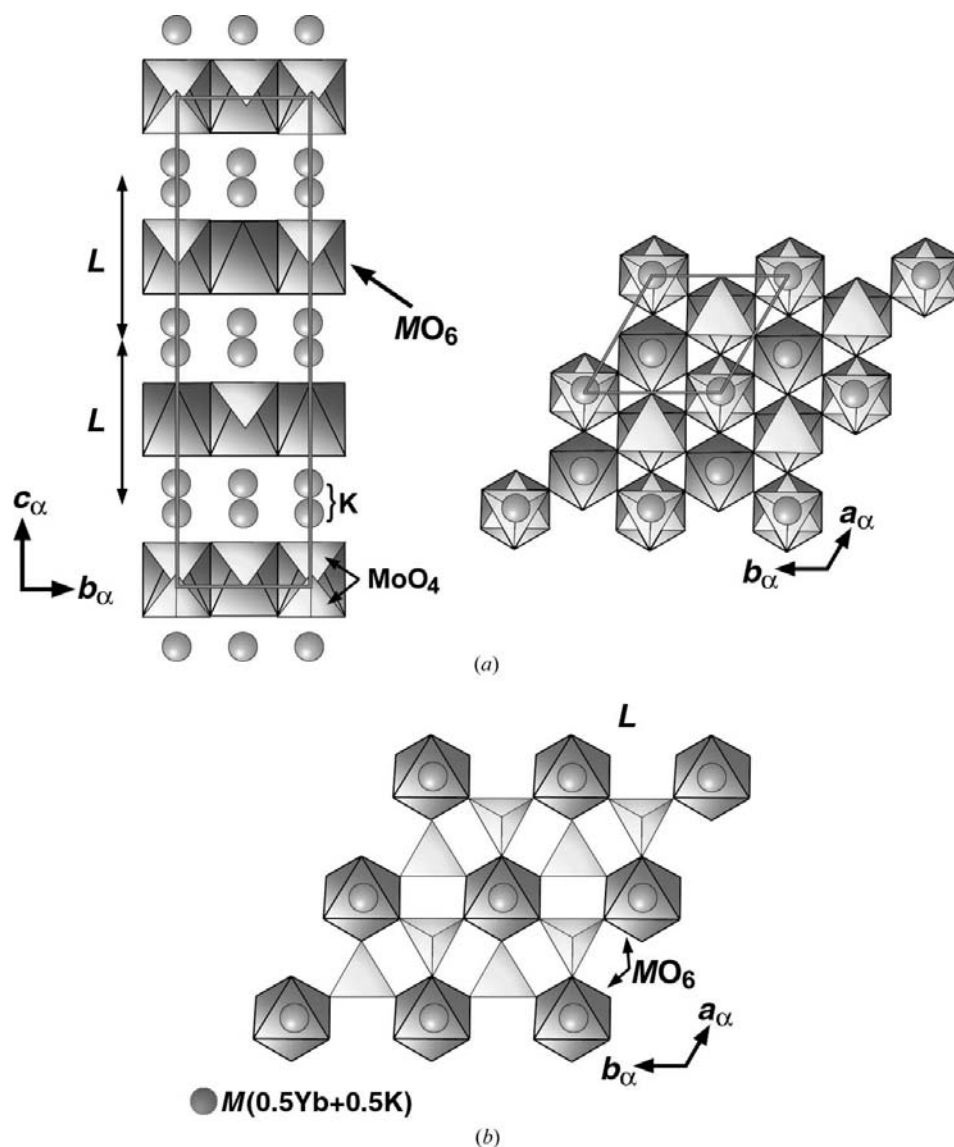


Figure 1

The palmierite structure type, space group $R\bar{3}m$. (a) Two projections of the structure. L layers are indicated. (b) A single L layer consisting of $[MoO_4]$ and $[MO_6]$ polyhedra, and K atoms which are located on both sides of the $[MO_6]$ octahedra along c .

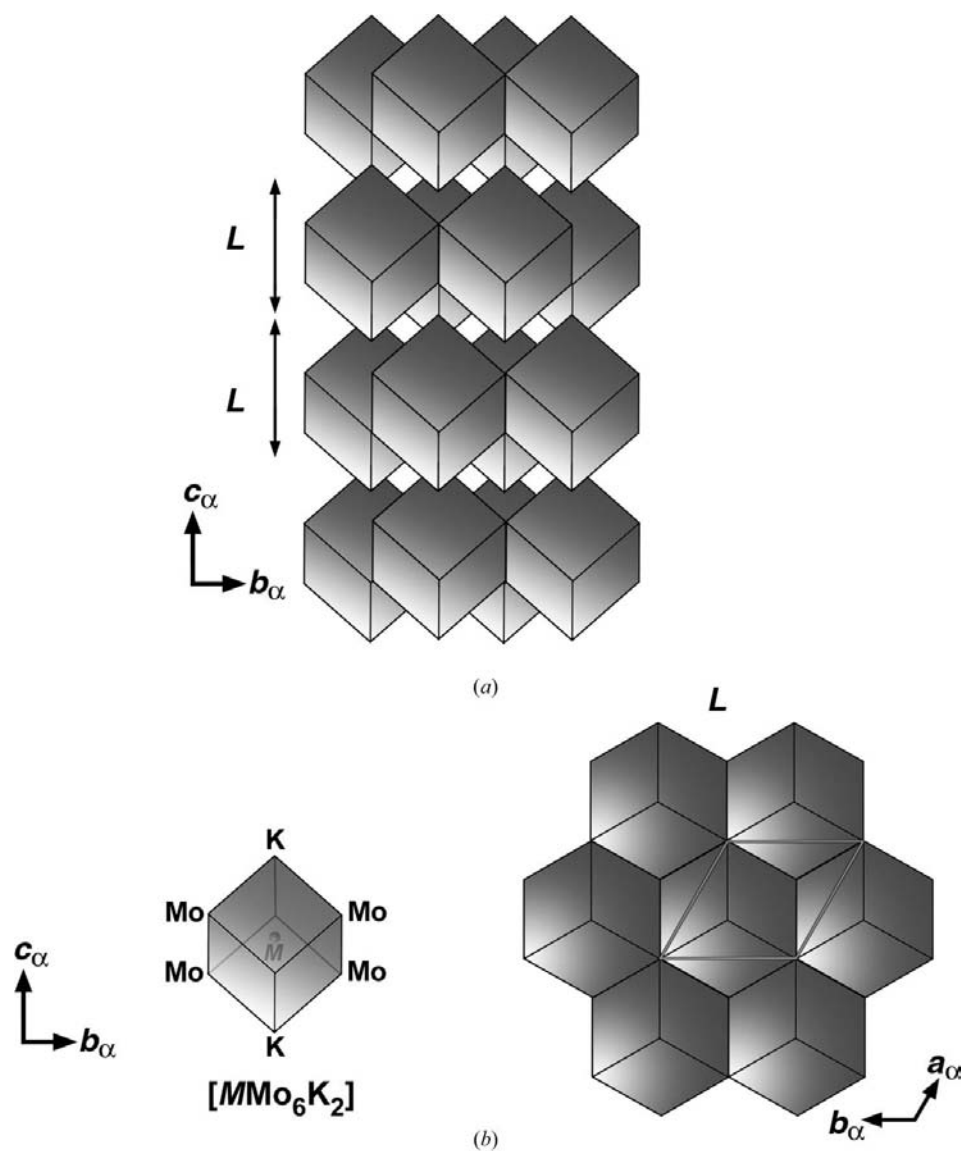


Figure 2
 The palmierite structure type as a combination of $[MMo_6K_2]$ cubic clusters. The projections in (a) and (b) (L) are analogous to Fig. 1. The structure of a single cluster is also represented.

reflections within the sphere limited by $\theta = 34.66^\circ$ including satellites up to third order were measured. The refinement of the unit-cell parameters, including the modulation vector \mathbf{q} , was performed using the *CrysAlis* software (Oxford Diffraction Ltd, 2001) and the program *NADA* (Schönleber *et al.*, 2001) in the conventional monoclinic setting and with $\mathbf{q} = [0\mathbf{a}^* + \beta\mathbf{b}^* + 0\mathbf{c}^*]$. The unit-cell parameters of the α , β and γ modifications are related by the following expressions: $\mathbf{a}_\beta = -\mathbf{c}_\gamma = (2\mathbf{a}_\alpha + \mathbf{b}_\alpha)$; $\mathbf{b}_\beta = 0.5\mathbf{b}_\gamma = \mathbf{b}_\alpha$; $\mathbf{c}_\beta = (\mathbf{a}_\gamma + 2\mathbf{c}_\gamma) = -4/3(2\mathbf{a}_\alpha + \mathbf{b}_\alpha) + 2/3\mathbf{c}_\alpha$. The non-primitive centring X in the $(3+1)$ -dimensional superspace (Table 1) corresponds to the three-dimensional average unit cell with $\mathbf{c}_{av} = 0.5\mathbf{c}_\beta$ and the space group $C2/m$.

The *JANA2000* system of programs (Petricek *et al.*, 2000) was used for structure refinement and analysis. The starting

models of the cations in the average structure were calculated from the rhombohedral α phase (Morozov *et al.*, 2003). This starting model can be represented by L layers of edge-sharing cubic clusters with the composition $[MMo_6K_2]$ (Fig. 2), which completely characterizes the palmierite (cationic) structure. The translation vector \mathbf{c}_α is related to the body diagonals of the cubes located at the centre of the R lattice, while \mathbf{a}_α is oriented along the face diagonal of the cube. The starting model was refined for the average structure in the monoclinic unit cell by using only the main reflections. The O atoms could not be unambiguously localized in the refined structure. By including first- and second-order satellites, it was possible to distinguish between the Yb and K1 atoms occupying the M position and their occupation could be reasonably described by two complementary crenel functions. The O atoms were located from a difference Fourier synthesis in the vicinity of Mo. The $[MoO_4]$ group was further refined as a rigid unit. The K-atom position of the α phase (Figs. 1 and 2) corresponds to K2 in the present structure. The position of K2 and the rigid-body groups (MoO_4) exhibit a clear discontinuity and their positional modulations can be divided into two mutually related t regions, which are best represented by crenel functions. The final coordinates, parameters of the relevant crenel functions and equivalent isotropic displacement parameters have been deposited.¹ Two positions of the $[MoO_4]$ rigid unit [(a) and (b) in the tables] are tilted and shifted, and the corresponding parameters are listed in Table 2.

A selection of relevant interatomic distances and angles are presented in Table 3 (a complete list of distances is available in the supplementary materials). The interatomic distances in the

supplementary materials are available in the supplementary materials.

¹ Supplementary data for this paper, including the Yb, K1, K2a and K2b Fourier components of the positional amplitudes, amplitudes of the displacive modulation functions, amplitudes for the rotation and translation modulations of the (a) and (b) $[MoO_4]$ rigid unit positions, displacement parameters and ADP modulation functions for all atoms, are available from the IUCr electronic archives (Reference: SN5021). Services for accessing these data are described at the back of the journal.

Table 1

Experimental data.

Crystal data	
Chemical formula	$K_5Yb(MoO_4)_4$
M_r	1008.3
System, superspace group	Monoclinic, $X2/m(0\beta 0)00$
Non-primitive translations	$[\frac{1}{2} \frac{1}{2} 0 0; 0 0 \frac{1}{2} \frac{1}{2}; \frac{1}{2} \frac{1}{2} \frac{1}{2} \frac{1}{2}]$
a, b, c (Å)	10.4054 (16), 6.1157 (12), 19.7751 (18)
β (°)	136.625 (10)
V (Å ³)	864.2 (3)
Z	2
D_x (Mg m ⁻³)	3.873 (1)
Modulation vector	$\mathbf{q} = 0.6354 (30)\mathbf{b}^*$
Radiation type	Mo $K\alpha$
No. of reflections for cell parameters	5412
θ range (°)	5.3–37.7
μ (mm ⁻¹)	9.46
Temperature (K)	293
Crystal form, colour	Isomorphic, colourless
Crystal size (mm)	0.14 × 0.13 × 0.12
Data collection	
Diffractometer	KM4
Data collection method CCD detector	KM4CCD/Sapphire
Absorption correction	Analytical
T_{min}	0.159
T_{max}	0.200
No. of measured, independent and observed reflections	67 857, 6142, 3412
Criterion for observed reflections	$I > 3\sigma(I)$
R_{int}	0.12
θ_{max} (°)	69.3
Range of h, k, l	$-12 \Rightarrow h \Rightarrow 12$ $-10 \Rightarrow k \Rightarrow 10$ $-35 \Rightarrow l \Rightarrow 35$
Refinement	
Refinement on	F
$R[F^2 > 2\sigma(F^2)], wR(F^2), S$	0.077, 0.085, 2.89
No. of reflections	3412
No. of parameters	94
Weighting scheme	$w = 1/[\sigma^2(F) + 0.0001F^2]$
$(\Delta/\sigma)_{max}$	0.001
$\Delta\rho_{max}, \Delta\rho_{min}$ (e Å ⁻³)	5.24, -5.20
Extinction method	B-C type 1 Lorentzian isotropic (Becker & Coppens, 1974)
Extinction coefficient	0.021 (6)

Computer programs used: *CrysAlis* (Oxford Diffraction Ltd, 2001).

first and second coordination spheres of Yb and K1 are represented as a function of the internal coordinate t in Fig. 3.

The length of the Yb crenel function [0.512 (2)] is slightly larger than the length of the K1 crenel function [0.488 (2)], while the stoichiometry of the compound, as confirmed by chemical analyses (Morozov *et al.*, 2003), requires equal amounts of Yb and K1 atoms. In order to explain this asymmetry, we tested both a model with a modulation of the Yb occupation and a model with an additional crenel function for a mixed ($K_{0.5}Yb_{0.5}$) composition. Both models led to unrealistic ADP parameters for the Yb and ($K_{0.5}Yb_{0.5}$) positions. Indeed, no essential modulations of the ADP parameters (see supplemental materials) were observed for the Yb position in our final model. Therefore, it is likely that over a short range along t ($\Delta t_{Yb} = 0.0248$) the modulation corresponds to a mixed

Table 2

Basic rotation and translation displacements for two (a and b) positions of the $[MoO_4]$ rigid unit in β - $K_5Yb(MoO_4)_4$.

Position	Rotation components (°)			Translation components		
	r_x	r_y	r_z	t_x	t_y	t_z
(a)	0	0	0	-0.00030 (11)	0	-0.00013 (6)
(b)	0	35.8 (5)	180	-0.0608 (5)	0	-0.0275 (3)

Table 3

Selected cation–oxygen distances (Å) and angles (°) in the $[MoO_4]$ tetrahedra in β - $K_5Yb(MoO_4)_4$.

	Average	Minimal	Maximal
Yb–O1a	2.293 (14)	2.168 (14)	2.686 (14)
Yb–O2a	2.31 (3)	2.21 (3)	2.45 (3)
Yb–O2b	2.26 (19)	2.2 (3)	2.36 (13)
K1–O1a	2.990 (17)	2.461 (18)	3.13 (2)
K1–O2a	2.84 (4)	2.55 (3)	3.10 (7)
K1–O1b	2.67 (3)	2.48 (3)	2.96 (3)
K1–O3a	2.97 (3)	2.90 (3)	3.17 (3)
K1–O3a	3.45 (4)	2.80 (8)	3.88 (4)
K2a–O1a	2.89 (3)	2.53 (3)	3.19 (2)
K2a–O1a	3.27 (4)	2.92 (4)	3.61 (4)
K2a–O1a	3.15 (3)	2.83 (3)	3.41 (3)
K2a–O2a	2.839 (15)	2.539 (18)	3.136 (15)
K2a–O2a	3.294 (15)	2.843 (18)	3.682 (17)
K2a–O3a	2.60 (3)	2.49 (4)	2.75 (3)
K2a–O1b	3.30 (6)	2.61 (5)	3.98 (5)
K2a–O1b	2.86 (6)	2.53 (6)	3.27 (4)
K2a–O3b	2.61 (18)	2.6 (3)	2.64 (13)
K2a–O3b	3.32 (4)	2.63 (4)	3.97 (3)
K2b–O1a	2.91 (7)	2.84 (9)	3.09 (6)
K2b–O1a	3.13 (8)	3.05 (5)	3.36 (6)
K2b–O2a	2.78 (6)	2.76 (9)	2.82 (4)
K2b–O2a	2.88 (2)	2.81 (2)	2.924 (18)
K2b–O1b	3.29 (13)	2.64 (9)	3.54 (10)
K2b–O2b	2.96 (8)	2.82 (6)	3.04 (12)
Mo a tetrahedron			
Mo–O1 × 2	1.78 (2)	1.73 (3)	1.83 (3)
Mo–O2	1.77 (3)	1.74 (3)	1.87 (4)
Mo–O3	1.72 (2)	1.684 (19)	1.77 (7)
O1–Mo–O2	110.4 (12)	108.2 (14)	113.3 (13)
O1–Mo a –O3	108.2 (11)	107.8 (10)	109.2 (11)
O1–Mo–O1	111.8 (18)	107.0 (17)	113.3 (17)
O2–Mo–O3	107.5 (11)	106 (2)	107.9 (10)
Mo b tetrahedron			
Mo–O1 × 2	1.77 (6)	1.72 (9)	1.85 (3)
Mo–O2	1.72 (16)	1.72 (13)	1.72 (11)
Mo–O3	1.68 (11)	1.63 (16)	1.76 (8)
O1–Mo b –O2	111 (4)	109 (4)	111 (5)
O1–Mo b –O3	109 (5)	108 (6)	110 (5)
O1–Mo b –O1	112 (8)	111 (9)	113 (6)
O2–Mo b –O3	106 (8)	105 (6)	106 (10)

position $M = (K_{0.5}Yb_{0.5})$. The Yb–O distances corresponding to the mixed t range ($\Delta t_{Yb} = 0.25 \pm 0.0062$ and 0.75 ± 0.0062 , Fig. 3) have unusually larger values, 2.4–2.68 Å, in comparison to the remaining part of the Yb t range, where the distances vary between 2.17 and 2.6 Å. Both Δt_{Yb} ranges are associated with the ($K_{0.5}Yb_{0.5}$) mixed composition (Fig. 3). In other words, approximately 2.5% of the M position have a mixed composition ($K_{0.5}Yb_{0.5}$). Both Yb and ($K_{0.5}Yb_{0.5}$) positions are characterized by the coordination number CN = 6.

2.3. Description of the incommensurate β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$ structure

Fragments of the incommensurately modulated β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$ structure related to the trigonal palmierite structure type (Fig. 1 and 2) are shown in Figs. 4 and 5.

Similar to the trigonal α phase (Fig. 1*a*), the β phase can be described as a series of L layers stacked along the c axis (Fig. 4). All L layers in the β phase also have identical compositions and topologies (Fig. 5). However, β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$ exhibits some characteristic features, which are related to two possible occupations of the M position (K1 or Yb). The distribution of

Yb and K1 in the structure is periodic along the superspace direction \mathbf{A}_2 , but aperiodic along the \mathbf{b} direction. Note that the two vectors are related by the expression $\mathbf{A}_2 = \mathbf{b} - (\mathbf{q}\cdot\mathbf{b})\mathbf{A}_4$ with $\|\mathbf{A}_4\| = 1$ (Fig. 6). Thus, the β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$ structure can be described by an aperiodic (but not arbitrarily) distribution of M clusters along the \mathbf{b} axis (Fig. 5).

3. Discussion

3.1. Complex cation–anion and cation–cation interactions

The basis of our analysis of the atomic interactions is the variation of interatomic distances along the internal coordinate t . If an independent variation is observed (*i.e.* not related by any symmetry operation), the strength of the interaction is inversely related to the magnitude of the variation.

The Mo–O distances and angles are the most stable entities in this structure (Table 3). As expected, the occupancy of the M position is correlated to the difference between the Yb–O and K1–O distances (Fig. 3). In spite of a stable coordination number, CN = 6, the Yb–O distances exhibit some large fluctuations with a maximal value of $\Delta d_{\text{max}} = 0.5 \text{ \AA}$ for a single Yb–O distance. The CN of K1 is 8, 10 or 12 depending on the value of the internal coordinate t . The K1–O distances vary between 2.46 and 3.88 \AA with $\Delta d_{\text{max}} = 1.08 \text{ \AA}$. For CN = 10, the K2–O distances vary more (from 2.49 to 3.98 \AA with $\Delta d_{\text{max}} = 1.37 \text{ \AA}$) than the K1–O distances. The variability of the distances related to the first coordination sphere of Yb and K atoms points to an additional type of interaction. This is related to the cation–cation interactions in the second coordination sphere. A comparison of the M –O and M –cation distances (Fig. 3) reveals that a larger number of shorter M –O distances are correlated to an increase in longer M –cation distances.

As seen in Figs. 3–5, the ordering of K1 and Yb in the M position induces a rotation and displacement of the MoO_4 tetrahedra in order to form YbO_6 octahedra. However, the Mo atoms at the centres of these units are stable,

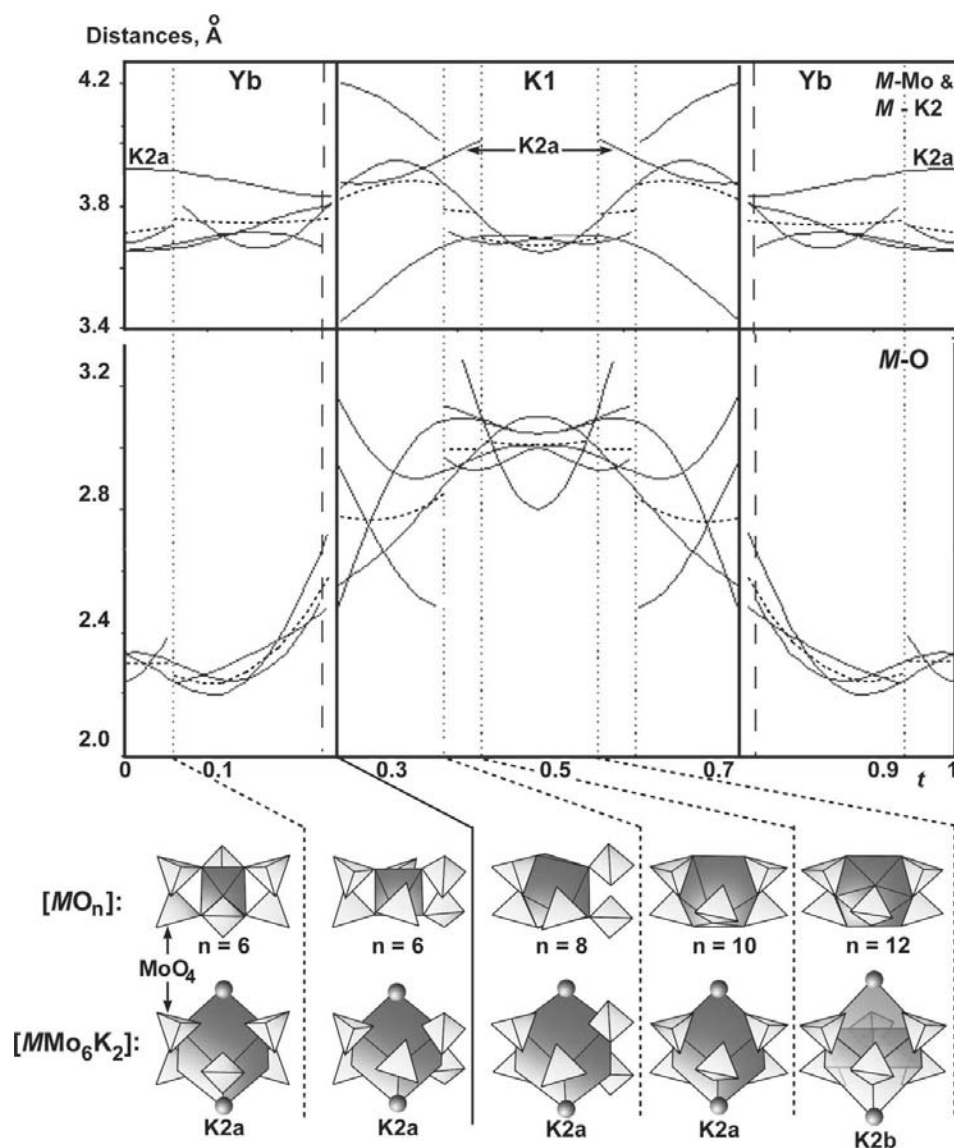


Figure 3 Interatomic distances in the second (upper part) and first coordination spheres (lower part) for the M atoms as a function of the internal coordinate t . In the upper part, only the M –K2 curves are indicated, with the remaining curves corresponding to M –Mo distances. The average distances are represented by dotted lines. Owing to the symmetry of the M position, every curve indicates two symmetrically equivalent distances. The t ranges related to $M = \text{K1}$ and $M = \text{Yb}$ are separated by solid vertical lines. The two t ranges located between the solid and dashed lines are associated with the mixed $M = [\text{K}_{0.5}\text{Yb}_{0.5}]$ composition. The vertical dotted lines separating the t ranges relate to different crenel functions. The corresponding shapes of the $[\text{MO}_n]$ and $[\text{MMo}_6\text{K}_2]$ polyhedra are also indicated.

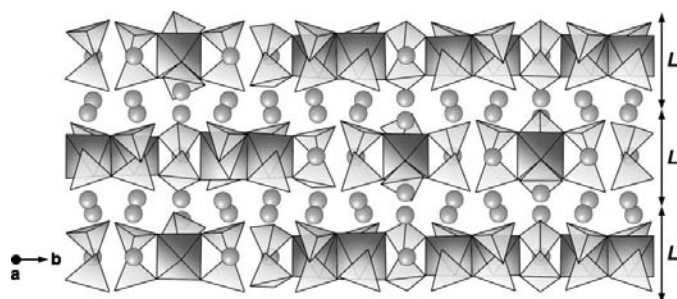


Figure 4
A fragment of the β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$ structure projection along the a axis. The YbO_6 octahedra (dark grey) and MoO_4 tetrahedra (light grey) are indicated along with the circles for the K atoms. The distribution of K and Yb is aperiodic along the b axis.

thus keeping the $[\text{MMo}_6\text{K}_{22}]$ cubic cluster unchanged. Since the Yb—Mo distances remain in the range 3.66–3.80 Å, the increase of the Yb—O distances in the octahedra is mainly

balanced by a shortening of the Yb—K2a distances from 3.92 to 3.82 Å in the Yb clusters (Fig. 3). Besides the MoO_4 tetrahedra, the Yb cluster has the least variable interactions ($\Delta d_{\text{max}} = 0.15$ Å).

The K1 clusters are less stable than the Yb cluster owing to the various coordination numbers in the first coordination sphere (Fig. 3). The shortest average K1—Mo distances in the second coordination sphere are correlated with the longest average K1—O distance in the first coordination sphere. The highest coordination number (CN = 12) of the K1—O interactions induces a transformation of the cubic K1 cluster into a $[\text{K1Mo}_6]$ octahedron as a consequence of the elongation of two K2b vertices (K1—K2b > 4.38 Å, Figs. 3 and 5). The balance between the K1—O and K1—K2 interactions can be directly associated with the close correlation between the K2 and MoO_4 crenel functions: in the lower part of Fig. 3 we observe that each K1—O coordination number is associated with a specific configuration of tilts and shifts of the MoO_4

tetrahedra; we also observe that the K1—K2 interactions vary for each coordination number. It is important to note that the third coordination sphere of K1 uniquely determines the CN in the first coordination sphere (Fig. 5): six Yb clusters surround a K1 [CN = 12] cluster; four Yb clusters and two K1 clusters surround a K1 [CN = 10] cluster; three Yb clusters and three K1 clusters surround a K1 [CN = 8] cluster.

The behaviour of the K2a atom can be better expressed by its contribution to the stabilization of the M cluster ($3.83 < \text{K2a}-M < 4.02$ Å with $\Delta d_{\text{max}} = 0.1$ Å, Fig. 3) than by its role as the centre of the K2aO_{10} polyhedron ($2.49 < \text{K2a}-\text{O} < 3.7$ Å with $\Delta d_{\text{max}} = 0.9$ Å). This contrasts with the K2b atom, which is not associated with an M cluster and which consequently exhibits a smaller variation of the K2bO_{10} polyhedron ($2.75 < \text{K2b}-\text{O} < 3.55$ Å and $\Delta d_{\text{max}} = 0.3$ Å). Three additional and relatively short K2—K2 distances (3.42–3.92 Å, and $\Delta d_{\text{max}} = 0.3$ Å) between non-shared vertices of the $[\text{MMo}_6\text{K}_{22}]$ cubic cluster should also be mentioned. These paired interactions contribute to the linking of two neighbouring L layers (Figs. 1 and 2). Here also the largest K2b—K2 distance, 3.8 Å, is smaller by 0.12 Å compared with the largest K2a—K2 distance, 3.92 Å, which is

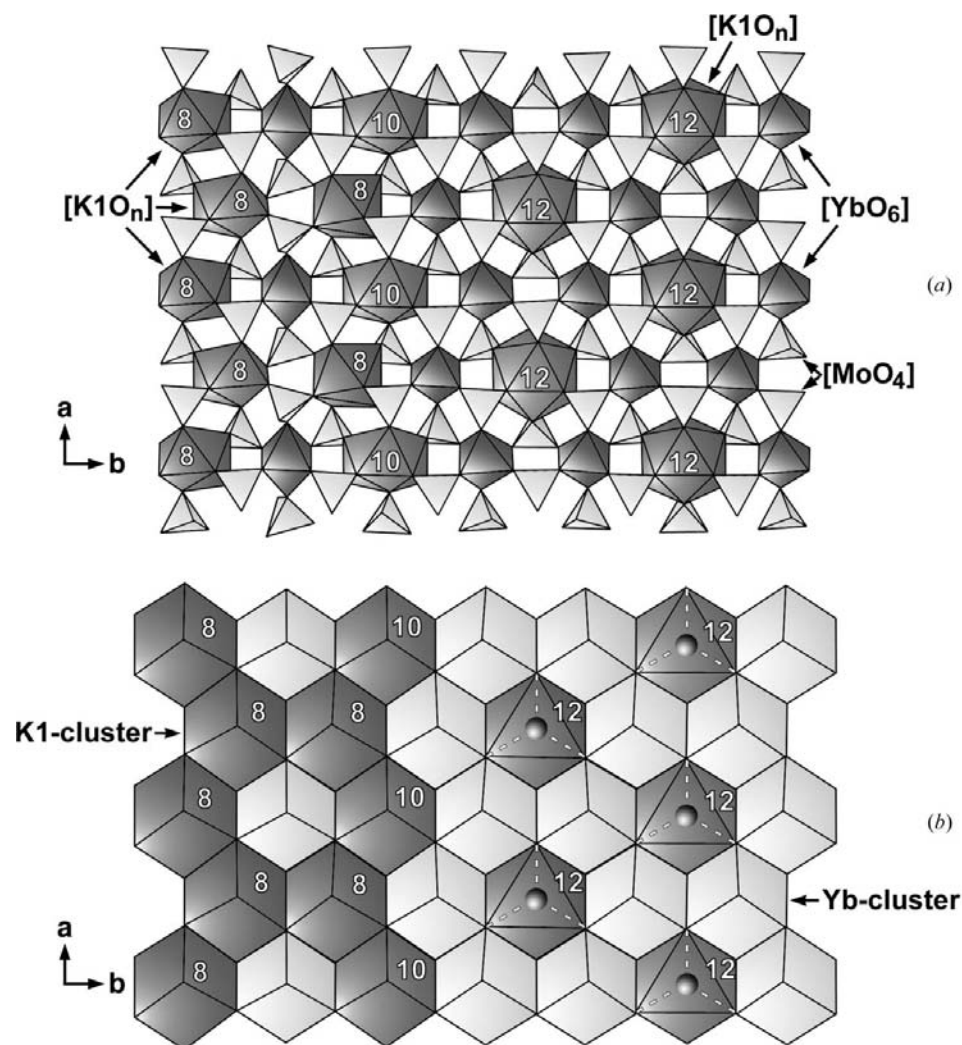


Figure 5
Fragment of a single L layer in β - $\text{K}_5\text{Yb}(\text{MoO}_4)_4$. The polyhedra of (a) the first and (b) the second coordination spheres are shown for $M = \text{Yb}$ and $M = \text{K1}$ atoms. The K2 atoms located on both sides of the M polyhedra along c are missing in (a). The $M = \text{Yb}$ and $M = \text{K1}$ clusters are shown in detail in Figs. 2 and 3. The numbers indicate the CN of K1 atoms in their first coordination sphere.

a consequence of the fact that K2b does not contribute to the formation of *M* clusters.

From the present analysis we can draw the following conclusions:

(i) Besides the MoO₄ tetrahedra, the cubic *M* clusters, [MMo₆K₂₂], are the most stable building units in the incommensurately modulated structure β-K₅Yb(MoO₄)₄ of the palmierite family. This observation justifies the interpretation of this structure in terms of building units of *M* clusters (Fig. 2).

(ii) The atomic interactions of the first (cation O atoms) and second (*M* cluster) coordination sphere are correlated.

(iii) The shape of the first coordination sphere of *M* = K1 atoms is uniquely determined by the third coordination sphere, *i.e.* it is directly related to the nature of the surrounding *M* clusters (Fig. 5*b*). In other words, the distribution of K1 and Yb atoms among the *M* clusters drives the tilts and shifts of the MoO₄ tetrahedra.

3.2. Driving forces of the modulation

The incommensurately modulated structure is associated with the distribution of K and Yb in the *M* position along the *b* axis. Therefore, the primary modulation parameter is the occupation of the *M* position, which is best described by complementary crenel functions for the K and Yb atoms. The distribution of the K1 and Yb clusters in the *L* layer determines the third coordination sphere of every K1 and Yb atom. The third coordination sphere of every K1 and Yb atom in

turn determines their first coordination sphere, *i.e.* the shift and tilt of the MoO₄ tetrahedra in order to form the bonds in the MO_{*n*} polyhedron (Fig. 5). Since the distribution of the K1 and Yb clusters is aperiodic along the modulation vector *q*, the tilt and shift of MoO₄ tetrahedra are modulated according to this distribution. The distances in the MO_{*n*} polyhedra (first coordination sphere) and in the *M* clusters (second coordination sphere) are correlated, thus leading to the modulation of the K2 atoms.

4. Conclusions

The second coordination sphere plays an important role in the formation of incommensurately modulated structures. This was predicted theoretically (Selke & Fischer, 1979; Janssen & Tjon, 1981; Bak, 1982; Parlinski & Chapuis, 1993) and the present study of β-K₅Yb(MoO₄)₄ is an additional example confirming this prediction. In the incommensurate structures of Na₂CO₃ and K₃In(PO₄)₂ (Chapuis & Arakcheeva, 2004; Arakcheeva *et al.*, 2003), and K₁₃P₄W₈O₃₂ (Dusek *et al.*, 2002) the origin of the modulation was also directly associated with interactions in the second coordination sphere, whereas in the palmierite-like structure β-K₅Yb(MoO₄)₄, these interactions affect the bonds between neighbouring atoms. The β-K₅Yb(MoO₄)₄ structure thus illustrates the complementary character of the interactions between the first and second coordination spheres. Moreover, this structure indicates that the third coordination sphere also plays an important role in the formation of modulated structures.

The contribution of the Swiss National Science foundation, grant No. 20-105325, is gratefully acknowledged. This work was also supported by the Grant Agency of the Czech Republic, grant 202/03/0430

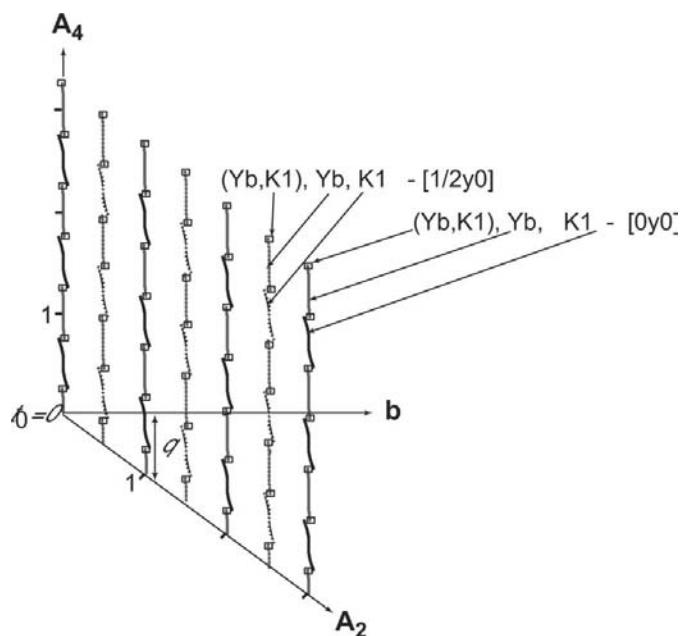


Figure 6 Occupational modulation of the *M* position in the incommensurate structure β-K₅Yb(MoO₄)₄, section *x*₂*x*₄ [*x*₃ = 0; *x*₁ = 0 and $\frac{1}{2}$]. Along *b*, the aperiodic sequence of K1 and Yb defines their distribution in the *L* layer (see the text, and Figs. 4 and 5): solid lines refer to *x*₁ = 0, whereas dashed lines refer to *x*₁ = $\frac{1}{2}$. The terms of the mixed (Yb,K) position are also indicated.

References

Arakcheeva, A., Chapuis, G., Petricek, V., Dusek, M. & Schönleber, A. (2003). *Acta Cryst.* **B59**, 17–27.
 Bak, P. (1982). *Rep. Prog. Phys.* **45**, 587–629.
 Becker, P. J. & Coppens, P. (1974). *Acta Cryst.* **A30**, 129–153.
 Chapuis, G. & Arakcheeva, A. (2004). *Z. Kristallogr.* **219**, 730–736.
 Dusek, M., Ludecke, J. & van Smaalen, S. (2002). *J. Mater. Chem.* **12**, 1408–1414.
 Janssen, T. & Tjon, J. A. (1981). *Phys. Rev. B*, **24**, 2245–2248.
 Morozov, V. A., Lazoryak, B. I., Lebedev, O. I., Amelinckx, S. & Van Tendeloo, G. (2003). *J. Solid State Chem.* **176**, 76–87.
 Oxford Diffraction Ltd (2001). *CrysAlis* Software System. Version 1.166. Oxford Diffraction Ltd, UK.
 Petricek, V., Dusek, M. & Palatinus, L. (2000). *JANA2000*. Institute of Physics, Praha, Czech Republic.
 Parlinski, K. & Chapuis, G. (1993). *Phys. Rev. B*, **47**, 13983–13991.
 Schönleber, A., Meyer, M. & Chapuis, G. (2001). *J. Appl. Cryst.* **34**, 777–779.
 Selke, W. & Fischer, M. E. (1979). *Phys. Rev. B*, **20**, 257–265.